

1 van der Waals Interactions on the Mesoscale: Open-Science 2 Implementation, Anisotropy, Retardation, and Solvent Effects

3 Daniel M. Dryden,[†] Jaime C. Hopkins,[#] Lin K. Denoyer,⁺ Lokendra Poudel,[¶] Nicole F. Steinmetz,^{†,‡,§,||}
4 Wai-Yim Ching,[¶] Rudolf Podgornik,^{#,•,^} Adrian Parsegian,[#] and Roger H. French^{*,†,||}

5 [†]Department of Materials Science and Engineering, [‡]Department of Biomedical Engineering, [§]Department of Radiology, School of
6 Medicine, ^{||}Department of Macromolecular Science and Engineering, ⁺Department of Physics, Case Western Reserve University,
7 10900 Euclid Avenue, Cleveland, Ohio 44106, United States

8 [#]Department of Physics, University of Massachusetts, Amherst, Massachusetts 01003, United States

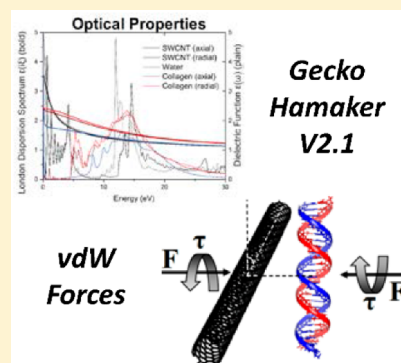
9 ⁺Deconvolution and Entropy Consulting, 755 Snyder Hill, Ithaca, New York 14850, United States

10 [¶]Department of Physics and Astronomy, University of Missouri—Kansas City, Kansas City, Missouri 64110, United States

11 [•]Department of Theoretical Physics, Jozef Stefan Institute, SI-1000 Ljubljana, Slovenia

12 [^]Department of Physics, Faculty of Mathematics and Physics, University of Ljubljana, SI-1000 Ljubljana, Slovenia

13 **ABSTRACT:** The self-assembly of heterogeneous mesoscale systems is mediated by
14 long-range interactions, including van der Waals forces. Diverse mesoscale
15 architectures, built of optically and morphologically anisotropic elements such as
16 DNA, collagen, single-walled carbon nanotubes, and inorganic materials, require a tool
17 to calculate the forces, torques, interaction energies, and Hamaker coefficients that
18 govern assembly in such systems. The mesoscale Lifshitz theory of van der Waals
19 interactions can accurately describe solvent and temperature effects, retardation, and
20 optically and morphologically anisotropic materials for cylindrical and planar
21 interaction geometries. The *Gecko Hamaker* open-science software implementation
22 of this theory enables new and sophisticated insights into the properties of important
23 organic/inorganic systems: interactions show an extended range of magnitudes and
24 retardation rates, DNA interactions show an imprint of base pair composition, certain
25 SWCNT interactions display retardation-dependent nonmonotonicity, and interactions
26 are mapped across a range of material systems in order to facilitate rational mesoscale
27 design.



1. INTRODUCTION

28 The understanding, design, and control of nanoscale and
29 mesoscale assembly presents a critical challenge across a range
30 of disciplines.^{1,2} Of particular interest for the nano-,³ soft-,⁴ and
31 biomatter communities⁵ are systems that are nanoscale in one
32 or two dimensions but macroscopic in the remaining
33 dimension(s),⁶ such as molecular wires and ribbons,⁷ single-
34 walled carbon nanotubes (SWCNTs),⁸ linear informational
35 macromolecules, fibrous proteins, and molecular sheets
36 including graphene,^{9,10} surfactant monolayers, and biological
37 membranes.¹¹ Universal in these systems are the long-range van
38 der Waals–London dispersion (vdW) interactions^{6,11} that arise
39 from dipolar fluctuations within optically contrasting objects in
40 an intervening medium. Existing theoretical approaches and the
41 computational implementations of these interactions can be
42 partitioned into the macroscopic (Lifshitz theory)¹¹ and
43 microscopic (few-atom) ab initio quantum chemical¹²
44 approaches.^{13–16} The microscopic approach is particularly
45 useful for atoms, molecules, and small clusters. The macro-
46 scopic approach is preferred for well-separated systems, where
47 the collective response of the matter can be approximated by a
48 frequency-dependent anisotropic dielectric function with sharp

spatial boundaries at a well-defined separation, as is the case in
49 systems with at least one macroscopic dimension. Under this
50 condition, the vdW interaction free energy is a functional of the
51 dielectric response $\epsilon''(i\xi)$ at discrete thermal (Matsubara)
52 frequencies on the imaginary frequency axis $i\xi$, which is itself a
53 functional of the imaginary part of the dielectric response
54 function $\epsilon''(\omega)$ calculated via the Kramers–Kronig rela-
55 tions.^{17,18} We have formulated a complete Lifshitz theory of
56 vdW interactions based on the idealized planar^{17,19,20} or
57 cylindrical morphology. This facilitates the calculation of vdW
58 interaction strengths for that interaction geometry, including
59 the optical anisotropy of interacting materials, and accounts for
60 retardation effects which arise from the finite propagation
61 velocity of electromagnetic disturbances, the defining feature of
62 the vdW interaction's separation dependence beyond the 10–
63 100 nm regime. The effect of the solvent and its own optical
64 dispersion can strongly influence interactions, even making
65 them nonmonotonic with attractive and repulsive separation
66

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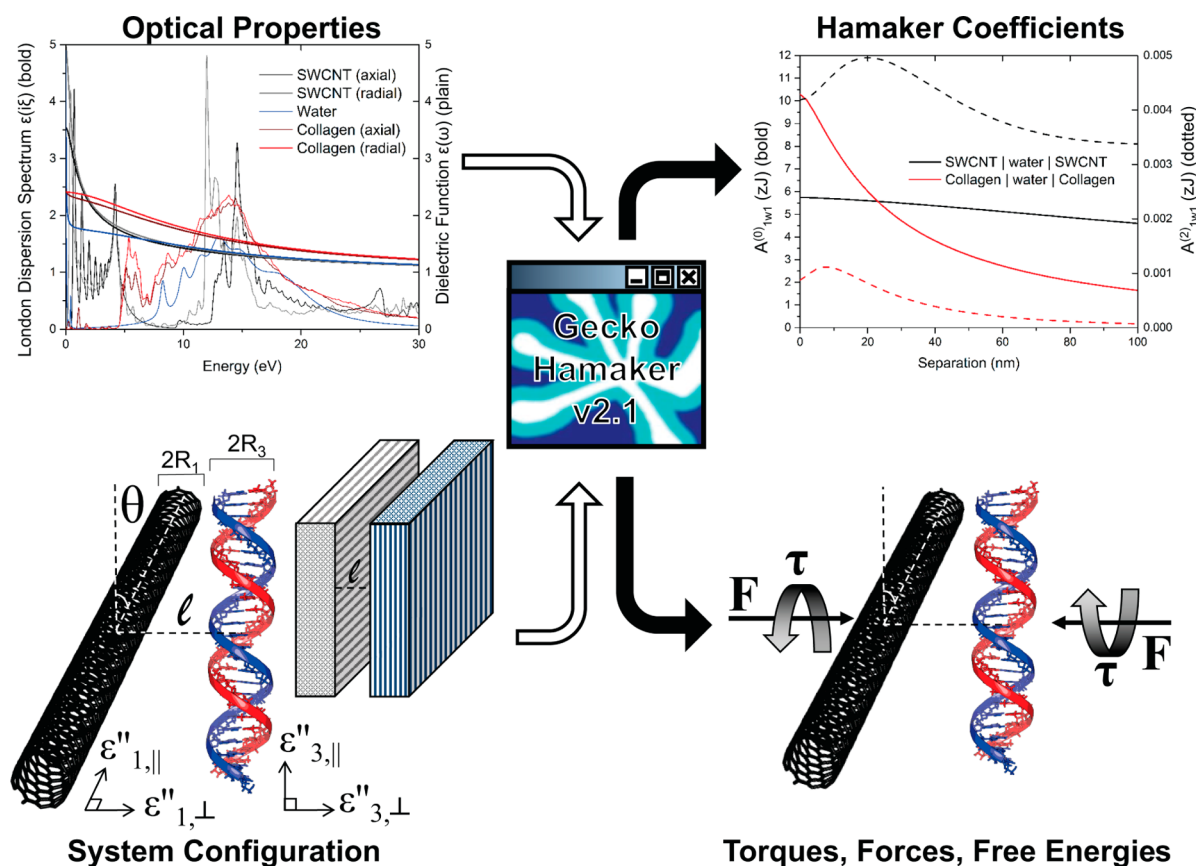


Figure 1. Schematic illustration of the *Gecko Hamaker* workflow using a large-radius (24,24,s) SWCNT and type I collagen in aqueous solvent as examples. Full spectral optical properties (upper left) are taken as input, experimentally measured using optical or electron energy-loss spectroscopy, or calculated using ab initio methods. An interaction geometry (lower left) is then chosen, consisting of optically anisotropic cylinders, optically anisotropic half-spaces, multilayered half-spaces, or graded-interface half-spaces. Specific optical properties from the spectral database are then applied to each element in the configuration. The outputs are the isotropic part $A^{(0)}_{1w3}$ (upper right, bold) and the anisotropic part $A^{(2)}_{1w3}$ of the Hamaker coefficients if applicable (upper right, dotted), as well as torques, normal forces, and thermodynamic interaction free energies (lower right). All quantities are given as functions of surface-to-surface separation l for a user-specified skew angle θ (for anisotropic systems) and radii R_1 and R_2 (for cylinders).

67 regimes.¹⁵ Optical input spectra may be obtained from
68 disparate methods, such as ab initio calculations or spectro-
69 scopic measurements.¹⁷

70 This formulation vastly improves upon previous isotropic¹⁷
71 and nonretarded⁸ results. We apply it to a range of
72 technologically promising cylindrical materials, including
73 metallic and semiconducting SWCNTs,^{8,21–24} multiple compo-
74 sites of DNA,^{25–27} collagen,²⁸ polystyrene,²⁹ and inorganic
75 materials. Our results are presented as Hamaker coefficients,
76 forces, and torques³⁰ for a wide range of symmetric (identical
77 materials interacting across an isotropic medium) and
78 asymmetric (different materials interacting across an isotropic
79 medium) interaction geometries in aqueous media. We
80 emphasize systems of general scientific interest, such as
81 organic-silica and DNA interactions, or those that exhibit
82 highly anomalous behavior, such as certain SWCNT
83 interactions. This detailed formulation of the Lifshitz theory
84 is implemented in the *Gecko Hamaker* open-source software
85 project³¹ and its optical property database (Figure 1). *Gecko*
86 *Hamaker* provides a powerful open-science tool for the
87 calculation of vdW interactions and serves the broader scientific
88 community as a versatile assembly thermodynamics and
89 mesoscale dynamics design tool whose applications are
90 augmented by an ever-expanding range of materials spectra
91 and interaction geometries. We demonstrate the versatility of

Gecko Hamaker as a design tool for the exploration and 92
discovery of novel features of long-range vdW interactions. We 93
demonstrate the capabilities of *Gecko Hamaker* to calculate 94
vdW interactions using a range of optically anisotropic cylinder 95
examples. 96

2. EXPERIMENTAL SECTION

2.1. Lifshitz Theory. From measured absorption spectra and/or ab 97
initio electronic structure calculations, we obtain the complex dielectric 98
function $\epsilon(\omega) = \epsilon'(\omega) + i\epsilon''(\omega)$, where the imaginary part 99
corresponds to energy dissipation in the material and via Kramers– 100
Kronig transform yields the London dispersion spectrum $\epsilon(i\xi_n)$, 101
describing the spontaneous field fluctuations at the origin of the vdW 102
force. In the Lifshitz theory the vdW interaction is a functional of 103
 $\epsilon(i\xi_n)$, evaluated at the discrete thermal Matsubara frequencies $\xi_n = 104$
 $2\pi n k_B T / \hbar$ ($n = 0, 1, \dots$), where k_B is the Boltzmann constant. At room 105
temperature, Matsubara frequencies are multiples of 2.4×10^{14} Hz or 106
0.025 eV. 107

To calculate the interaction between pairs of anisotropic cylinders 108
of materials 1 and 3, we modify the simple case of two uniaxially 109
anisotropic planar half-spaces acting across an isotropic medium¹⁹ by 110
imagining the material to be composed of arrays of parallel cylinders 111
with different anisotropic polarizabilities^{8,32} and then extracting the 112
pair interaction between uniaxial anisotropic cylinders of radii R_1 and 113
 R_3 via the Pitaevskii method.¹¹ The intervening space between the two 114
arrays and the space between cylinders is assumed to be filled with an 115

116 optically isotropic aqueous solvent of $\epsilon_m(\omega)$ described by a single-
117 Debye and multiple-Lorentz oscillator model.¹¹

118 For $\epsilon_{\parallel}^{1,3}(\omega)$, the dielectric response parallel to the longitudinal axis
119 of the cylinder, and $\epsilon_{\perp}^{1,3}(\omega)$, the dielectric response perpendicular to
120 the longitudinal axis of the cylinder, we can define the relative
121 anisotropy measures $\Delta_{\parallel}^{1,3}(\omega)$ and $\Delta_{\perp}^{1,3}(\omega)$ as

$$\Delta_{\parallel}^{1,3}(\omega) = \frac{\epsilon_{\parallel}^{1,3}(\omega) - \epsilon_m(\omega)}{\epsilon_m(\omega)}, \quad \Delta_{\perp}^{1,3}(\omega) = \frac{\epsilon_{\perp}^{1,3}(\omega) - \epsilon_m(\omega)}{\epsilon_{\parallel}^{1,3}(\omega) - \epsilon_m(\omega)} \quad (1)$$

122
123 The vdW interaction free energy between two semi-infinite anisotropic
124 uniaxial dielectric layers across a finite layer of thickness l was worked
125 out in the nonretarded limit¹¹ and the fully retarded limit¹⁹ as a
126 function of their separation l and the angle between their principal
127 dielectric anisotropy axes θ : $G(l, \theta)$. It then follows that the interaction
128 free energy between two cylinders, $G(l, \theta)$, whose axes are contained
129 within the two parallel boundaries at a separation l but skewed at an
130 angle θ is given by the second derivative $d^2G(l, \theta)/dl^2$ expanded to
131 second order in the density of the two cylindrical arrays.¹¹ Note that
132 such an expansion is possible only if the dielectric response at all
133 frequencies is bounded so that in the case of materials with free
134 charges one needs to model them explicitly.³³ vdW interactions
135 between optically anisotropic cylinders acting across an isotropic
136 medium create torques, given by

$$\tau = -\frac{dG(l, \theta)}{d\theta}$$

137 as well as attractions or repulsions in the normal force

$$F = -\frac{dG(l, \theta)}{dl}$$

138 As the separation between the bodies decreases, they feel mutual
139 torques that favor the alignment of the principal axes. The free energy
140 has a periodic dependence on the mutual angle, but the limit as $\theta \rightarrow 0$
141 has to be considered carefully as in the parallel configuration the free
142 energy scales with the length of the cylinders.³²

143 The free energy of the vdW interactions between two skewed
144 cylinders is then obtained as

$$G_{1m3}(l, \theta) = -\frac{(\pi R_1^2)(\pi R_3^2)}{2\pi l^4 \sin \theta} (A_{1m3}^0(l) + A_{1m3}^2(l) \cos 2\theta) \quad (2)$$

146 where the inverse $\sin \theta$ dependence stems from the shape
147 (morphological) anisotropy and the $\cos 2\theta$ dependence stems from
148 the material anisotropy. For both symmetric and asymmetric systems
149 with cylindrical morphology, Hamaker coefficients $A^{(0)}$ and $A^{(2)}$ are
150 functions of separation l (Figure 1) and the ratio of relative anisotropy
151 measures $a_{1,3}(i\xi_n)$, which depend on the material types and sampling
152 frequencies

$$a_{1,3}(i\xi) = \frac{2\Delta_{\perp}^{1,3}(i\xi)}{\Delta_{\parallel}^{1,3}(i\xi)} \quad (3)$$

154 Both Hamaker coefficients are defined through a summation over the
155 Matsubara frequencies ξ_n :

$$A_{1m3}^0(l) = \frac{k_B T}{32} \sum_{n=0}^{\infty} \Delta_{1,\parallel} \Delta_{3,\parallel} p_n^4(l) \int_0^{\infty} t dt \frac{e^{-2p_n(l)\sqrt{t^2+1}}}{t^2+1} \tilde{g}^{(0)}(t, a_1(i\xi_n), a_3(i\xi_n)) \quad (4)$$

157 with

$$\tilde{g}^{(0)}(t, a_1, a_3) = 2[(1+3a_1)t^2 + (1+a_1)][1 \rightarrow 3] + 4(1+a_1)(1+a_3)$$

158 and

$$A_{1m3}^2(l) = \frac{k_B T}{32} \sum_{n=0}^{\infty} \Delta_{1,\parallel} \Delta_{3,\parallel} p_n^4(l) \int_0^{\infty} t dt \frac{e^{-2p_n(l)\sqrt{t^2+1}}}{t^2+1} \tilde{g}^{(2)}(t, a_1(i\xi_n), a_3(i\xi_n), \theta) \quad (5)$$

with

$$\tilde{g}^{(0)}(t, a_1, a_3, \theta) = (1-a_1)(1-a_3)(t^2+2)^2$$

The dimensionless factor $p_n(l) = \epsilon_m(i\xi_n)((\xi_n^2)/(c^2))l^2$, which depends
161 on the ratio of the travel time of light, l/c , across the intervening
162 separation and the fluctuation lifetime, $1/\xi_n$, captures the separation
163 dependence and retardation due to the finite speed of light. For the
164 zero-frequency Matsubara term, the lifetime of the fluctuations is
165 sufficiently long not to be affected by retardation.

166 Alternative methods for evaluating the strength of vdW interactions
167 also exist and are robust under certain conditions. Density functional
168 theory (DFT)-based methods^{12,34} are standardly useful for atoms and
169 small clusters, but they usually become unwieldy for systems with large
170 numbers of electrons. However, recent advances based on fluctuating
171 dipoles that capture both intra- and intermolecular collective
172 fluctuations^{35,36} are indeed also applicable to larger systems.
173 Nevertheless, it is important to appreciate that the solvent effect,
174 specifically the effect of the aqueous solvent, is crucial to many
175 applications of vdW interactions in soft- and biomatter contexts. While
176 the solvent effects and the related large contribution of the zero-
177 frequency Matsubara (classical) term to the total vdW interactions
178 both enter the Lifshitz theory “naturally” and on the same level as the
179 fluctuation response of the interacting matter,¹⁸ it might be quite
180 difficult to implement them by more microscopic methods. This is the
181 main reason that the Lifshitz theory retains its relevance in particular
182 for soft matter- and biomatter-related problems.

183 For larger objects or equivalently small separations, the proximity
184 force approximation (PFA), also referred to as the Derjaguin
185 approximation,^{15,37,38} can accurately describe objects such as colloids
186 on the order of micrometers in radius, provided the separation
187 between the objects is much smaller than the characteristic dimensions
188 of the objects. However, this regime is of limited applicability for
189 nanoscale objects where the necessary object separations would be
190 subangstrom. The mesoscale Lifshitz theory described here in its large-
191 separation regime is valid for mesoscale objects with separations on the
192 nanometer scale, making it well-suited for describing realistic nano-
193 and mesoscale interactions.

The Lifshitz theory is based on the dielectric response function and
195 can be formulated for all the cases where this response function
196 exists.¹¹ It is valid for the macroscale, mesoscale, and microscale: in
197 fact, even the atomic pair potentials clearly follow from the application
198 of the Pitaevskii ansatz to the general Lifshitz formula for the
199 interaction between macroscopic bodies.³³ On this level, the question
200 of collective fluctuations for interacting many-body aggregates enters
201 the Lifshitz theory solely through the dielectric response function and
202 can be fully analyzed on that level. The calculation of the latter is
203 therefore technically not a part of the Lifshitz theory per se but has to
204 be imported from a separate full many-body theory of the dispersion
205 spectra. The clear decomposition into the calculation of the spectral
206 properties and the consequent long-range vdW interaction is the
207 principal feature of the Lifshitz theory. Thus, it is not the Lifshitz
208 theory itself that needs to be compared or counterposed to other more
209 microscopic theories of vdW interactions but the methodology of
210 getting the appropriate dielectric response functions (dispersion
211 spectra). If the full many-body dielectric response function were
212 known in its entire time and space domain, then the Lifshitz theory
213 would in principle be able to provide a complete and consistent
214 description of the vdW interactions.

215
216 **2.2. Gecko Hamaker.** Recent advances in the theory of vdW
217 interactions have facilitated the implementation of an open-source,
218 open-data design tool that facilitates an understanding and prediction
219 of the magnitude and properties of vdW interactions in a wide variety
220 of contexts. The *Gecko Hamaker* open-source software project is a full
221 implementation of the mesoscale classical Lifshitz theory for isotropic
222 and anisotropic planar multilayer,¹⁷ sharp, or graded interfaces for

223 modeling grain boundaries³⁹ and cylinder–cylinder³² interaction
 224 geometry with an intervening dielectric medium, accompanied by an
 225 extensive database of material optical properties spectra. The machine-
 226 readable optical properties database is available either for download or
 227 as a web service and makes available the full spectral properties of over
 228 100 materials from both ab initio calculations^{3,40} and experi-
 229 ments,^{6,39,41} including inorganic as well as organic materials such as
 230 type I collagen and (GC)10 duplex DNA spectra.^{40,42} The optical
 231 properties included in the database may be applied to both macro- and
 232 nanoscale objects that are macroscopic in at least one dimension such
 233 as cylinders and sheets. It is critical to note that the objects described
 234 in *Gecko Hamaker* are macroscopic in at least one dimension and
 235 furthermore that nanoscale-sized probes have in certain cases³⁹ been
 236 used to directly measure nanoscale optical properties, producing
 237 consistent results as inputs to the classical mesoscale Lifshitz theory.
 238 The *Gecko Hamaker* software and its source code are distributed freely
 239 on Sourceforge³¹ under the GNU general public license (gnu.org).
 240 This open-science architecture is at the forefront of a growing trend,
 241 with both the U.S. government^{43,44} and the G8 nations⁴⁵ emphasizing
 242 data sharing for advancing the dialogue of scientific discovery. Existing
 243 programs such as *Scuff-EM*⁴⁶ provide useful and versatile implementa-
 244 tions for the Casimir community⁶ based on user-specified geometries
 245 at zero temperature and optically isotropic systems in vacuum. *Gecko*
 246 *Hamaker* uses predefined geometries but allows for the calculation of
 247 temperature dependence, optical anisotropy, and variable media and
 248 provides both an existing database of material properties and a
 249 graphical user interface to increase its accessibility even to non-
 250 specialists. By making the *Gecko Hamaker* open data science tool with
 251 its spectra database freely available, we increase the opportunities for
 252 collaboration between researchers focused on long-range interactions
 253 and mesoscale design engineers in energy materials, geology,
 254 chemistry, physics, and biology.

3. RESULTS AND DISCUSSION

255 vdW interaction properties between different optically aniso-
 256 tropic materials in cylindrical interaction geometry are
 257 presented as Hamaker coefficients, normal force, and torque
 258 (Experimental Section). The anisotropic dielectric response of
 259 the materials involved introduces two distinct Hamaker
 260 coefficients: the isotropic part, $A_{1w1}^{(0)}$, and the anisotropic
 261 part, $A_{1w1}^{(2)}$. Apart from the anisotropy of the dielectric
 262 response, the interaction free energy also contains the effects of
 263 anisotropic morphology and interaction geometry.

264 Within the Lifshitz formalism, the interaction free energy
 265 depends on the London dispersion spectrum $\epsilon''(i\xi_n)$, discretely
 266 sampled at thermal Matsubara frequencies ξ_n obtained from
 267 the Kramers–Kronig transform of $\epsilon''(\omega)$. The latter is an input
 268 function and can be acquired by a variety of experimental and
 269 theoretical methods.¹⁷ The *Gecko Hamaker* software platform
 270 already contains an extensive spectral database. We specifically
 271 used a combination of experimental data fits (for water)¹¹ and
 272 the ab initio orthogonalized linear combination of atomic
 273 orbital (OLCAO) method, explained elsewhere,³⁵ to calculate
 274 the electronic structure and optical properties of interacting
 275 materials. Hamaker coefficients obtained by the ab initio optical
 276 spectra for a material may sometimes differ from those
 277 calculated from an experimental spectrum of the same material.
 278 Optical contrast at different energies contributes nonuniformly
 279 to the strength of the vdW interaction;¹⁸ as such, the bandgap
 280 underestimation that results from the local density approx-
 281 imation of density functional theory calculations⁴² tends to
 282 skew the Hamaker coefficients. The details of this effect and
 283 strategies for its mitigation are a topic of ongoing research.¹⁸
 284 Apart from this caveat, it is nevertheless interesting to compare
 285 trending behavior across different methods. All interactions are
 286 evaluated in an aqueous medium (denoted by w) with the zero-

frequency (static) Matsubara term completely screened,¹¹ 287
 assuming that the free charge carrier (salt ion) concentration 288
 in the bathing medium is sufficiently high that the zero- 289
 frequency Matsubara contribution to $A_{1w1}^{(0)}$ and $A_{1w1}^{(2)}$ 290
 vanishes (Experimental Section). This condition is met at 291
 separations of 1 nm or greater, whenever the ion concentration 292
 in the intervening medium is greater than $\sim 10^{20}$ cm⁻³, 293
 equivalent to a univalent ion solution concentration of ~ 50 294
 mM. This is a reasonable assumption for a wide range of 295
 realistic conditions, including the commonly used buffers in 296
 biological samples. The unscreened zero-frequency Matsubara 297
 term may be large in certain systems, especially those with high 298
 optical anisotropy at low frequencies, e.g., all systems involving 299
 an aqueous medium but also metallic SWCNTs, where the 300
 chiral indices⁸ (n, m) are such that $(n - m)/3 = \text{integer}$. For 301
 amorphous SiO₂, the zero-frequency contribution to $A_{1w1}^{(0)}$ is 302
 2.15 zJ; by comparison, for the (5,2,m) metallic SWCNT it is 303
 50.5 zJ. This screening effect may be most significant for 304
 materials with a small dielectric contrast at all energies. For 305
 these materials, screening out the relatively large zero-frequency 306
 contribution can result in a large change in the free energy as 307
 well as increase the weighting of contributions from medium 308
 and high energies. Here we observe that the static contribution 309
 for the (5,2,m) SWCNT, while large, accounts for only 4.4% of 310
 the interaction strength in that system; for aSiO₂, the zero- 311
 frequency term accounts for 24.8% of the interaction strength. 312
 For idealized metallic systems in vacuo, without any screening 313
 of the zero-frequency term, the low-temperature vdW 314
 interactions would converge to Casimir forces,¹³ a case we do 315
 not analyze here.¹⁵ Wherever the isotropic and anisotropic 316
 Hamaker coefficients are discussed as single values, the surface- 317
 to-surface separation between idealized cylinders is taken to be 318
 5 nm. At this spacing, retardation can have a noticeable effect 319
 on the high energy contributions to the summation terms of 320
 the Hamaker coefficients, but the low-energy contributions 321
 tend to remain unaffected. 322

3.1. Range of Magnitudes. The $A_{1w1}^{(0)}$ and the 323
 $A_{1w1}^{(2)}$ Hamaker coefficients for cylindrical bodies may 324
 take on a wide range of values in symmetric systems (denoted 325
 1w1), where both cylinders consist of the same material (Figure 326
 2). The largest $A_{1w1}^{(0)}$ value seen are for metallic SWCNTs, 327

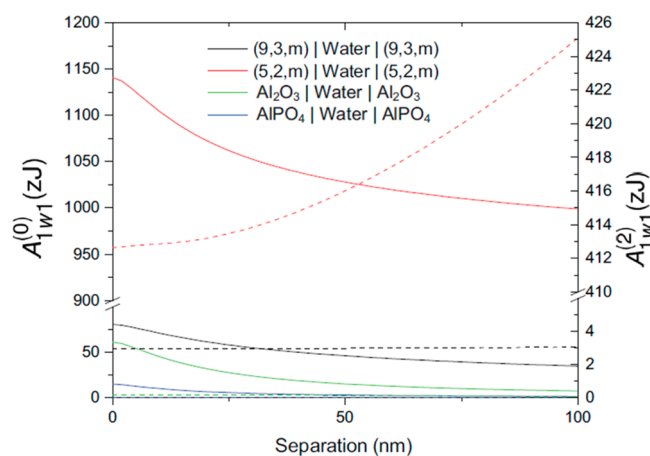


Figure 2. Range of magnitudes for symmetric configuration $A_{1w1}^{(0)}$ isotropic (solid lines) and anisotropic $A_{1w1}^{(2)}$ (dotted lines) cylindrical Hamaker coefficients for the fully retarded formulation, with the zero-frequency contribution screened in the intervening aqueous medium.

including the chiralities (5,2,m) and (9,3,m). Hamaker coefficients for SWCNTs are consistent with those published previously,⁸ and (5,2,m), because of its high optical anisotropy, exhibits an $A_{1wl}^{(0)}(l)$ value an order of magnitude larger than any other material studied here. Inorganic ceramics such as Al_2O_3 and AlPO_4 but also fibrous proteins such as type I collagen tend to exhibit substantial Hamaker coefficients. In general, materials with a lower optical contrast with water tend to exhibit smaller Hamaker coefficients, as observed for DNA, large-radius SWCNTs such as (24,24,s), aSiO_2 , and polystyrene (Figure 3). While Hamaker coefficients for various types of

coefficients as a function of separation, as observed previously in other systems.^{32,47}

3.3. Angular Dependence. The vdW interaction free energy depends on the mutual angle between interacting cylinders in two ways (Experimental Section): the anisotropic shape of the cylinders and the anisotropic dielectric response of the cylinder material. Because of the extreme anisotropy of the cylindrical shape, the interaction free energy shows an overall $1/\sin(\theta)$ dependence on the mutual orientation of their axes as they rotate from an aligned to a perpendicular configuration. In addition, anisotropies in material dielectric responses lead to two Hamaker coefficients, $A^{(0)}$ and $A^{(2)}$, with the second one weighted by a $\cos(2\theta)$ angular dependence. In general, the magnitude of $A^{(0)}$ is much larger than that of $A^{(2)}$, so $A^{(0)}$ dominates the angular dependence and sign of the free energy and torque. Anisotropic Hamaker coefficient $A_{1wl}^{(2)}$ for cylindrical morphology, which describes the dependence of the Hamaker coefficient on the skew angle θ between the cylinders, shows an unusual nonmonotonic dependence on separation (Figures 2 and 3). Unlike $A_{1wl}^{(0)}$, which in general tends to decrease monotonically, $A_{1wl}^{(2)}$ exhibits a pronounced local maximum at separations ranging from a few nanometers to a few hundred nanometers, suggesting that vdW interactions in some systems may have a more pronounced angular dependence within a certain regime of separations. In general, $A_{1wl}^{(0)}$ has a complex dependence on the optical (parallel and perpendicular response) anisotropy in the system. The nonmonotonic retardation effect is seen even in materials with no optical anisotropy, e.g., aSiO_2 , indicating the geometric anisotropy as its primary source; the magnitude of the effect and the position of the maximum of the anisotropic Hamaker coefficient are nonetheless modified by the particular nature of the material's optical properties and optical anisotropy. As with the $A_{1wl}^{(0)}$, $A_{1wl}^{(2)}$ also varies drastically for different materials, with a strong correlation with the degree of optical anisotropy in the material. (AT)10 DNA has an $A_{1wl}^{(2)}$ that is effectively zero, even at small separations, indicating that the contribution of optical anisotropy may be quite small in such systems. Conversely, metallic SWCNTs have a large optical anisotropy and subsequently exhibit anomalously high $A_{1wl}^{(2)}$ values, as large as 33.6 zJ for (5,2,m) SWCNTs (Figure 3). Even the (24,24,s) SWCNT, which has a relatively modest value of $A_{1wl}^{(0)}$ at small separations, has an $A_{1wl}^{(2)}$ that is an order of magnitude larger than for less-anisotropic materials with comparable $A_{1wl}^{(0)}$ values (Figure 3).

3.4. Asymmetric Systems. Understanding the interactions in asymmetric systems, where materials 1 and 3 differ, is paramount for mesoscale system design. A particularly relevant case for applications in sensors⁴⁸ and nanobio interfaces⁵ is the vdW interaction of biomaterials with silica nanorods (Figure 4). Silica's interaction with collagen results in a Hamaker coefficient that is 39% larger than that of the silica-(GC)10 DNA interaction at 5 nm separation. However, both interactions are strongly diminished by retardation effects compared to the silica-(24,24,s) SWCNT interaction. This phenomenon is tied to the details of the particular optical spectra that in turn govern the details of the retardation screening. The values of $A_{1wl}^{(2)}(l)$ clearly exhibit a more complex nonmonotonicity than that observed in symmetric systems; this effect is discussed in detail below. In some cases (not shown here), the dielectric responses of materials allow for a repulsive interaction, i.e., a negative Hamaker coefficient.^{32,47} Consequently, this implies that torques also have a sign

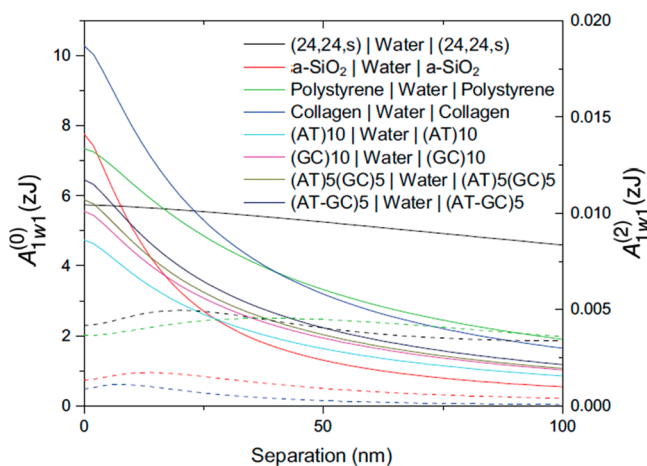


Figure 3. Symmetric configuration isotropic $A_{1wl}^{(0)}$ (solid lines) and anisotropic $A_{1wl}^{(2)}$ (dotted lines) Hamaker coefficients for cylindrical morphology in the fully retarded formulation with a screened zero-frequency contribution in the case of an intervening aqueous medium. All materials here, including all biomaterials included in this study, have $A_{1wl}^{(0)} < 10$ zJ.

DNA molecules are comparatively smaller, they do depend on the base-pair sequence details and could control the finer details of the equilibrium assembly structure.

3.2. Retardation Effects. The effect of retardation is seen in the rate of change of $A_{1wl}^{(0)}(l)$ with separation. It is strongly material-dependent and is due to the finite speed of light across the intervening medium, dephasing the correlated fluctuations with small lifetimes, i.e., high frequencies. The high-frequency contributions to the optical contrast in a system vary with the particular London dispersion spectrum $\epsilon(i\xi)$ for each material.¹¹ If the vdW interaction within a particular system results primarily from high-frequency contributions, such a system will exhibit a faster rate of retardation than a system where the interaction results from lower-frequency optical contrast. The (24,24,s) SWCNT (Figure 3, black) does not have a particularly large $A_{1wl}^{(0)}(l)$ at small separations, but it exhibits little retardation. Thus, its Hamaker coefficient for ~ 50 nm becomes by far the largest in this set. The majority of its optical contrast with water occurs at low frequencies; it is thus relatively impervious to retardation (optical properties inset, Figure 1). In contrast, type I collagen (Figure 3, blue) has its optical contrast with water spread out over a wide range of frequencies, including a significant contribution from high-energy Matsubara frequencies. Because of this, it exhibits dramatic retardation because these frequencies rapidly dephase with increasing separation. Under certain conditions, the effects of retardation can lead to a change of sign in the Hamaker

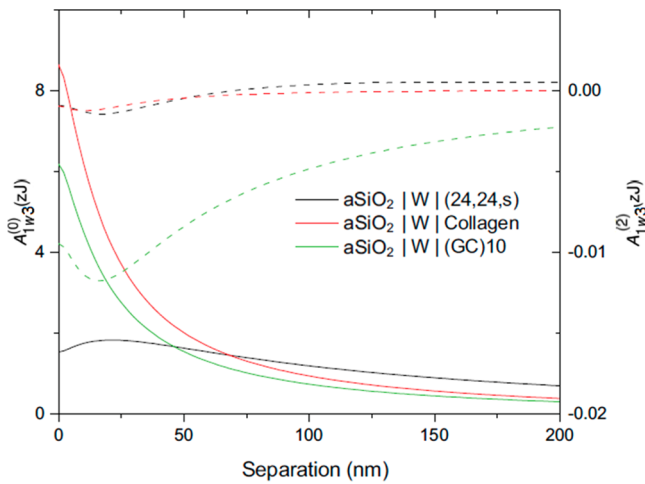


Figure 4. Asymmetric configuration $A_{1w3}^{(0)}$ (solid lines) and $A_{1w3}^{(2)}$ (dotted lines) cylindrical Hamaker coefficients for the interaction of silica nanorods with biological materials in the fully retarded formulation with a screened zero-frequency contribution for an intervening aqueous medium.

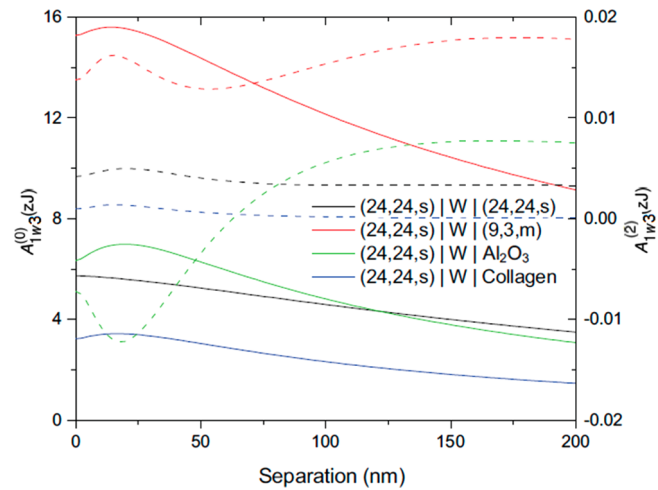


Figure 5. Asymmetric configuration $A_{1w3}^{(0)}$ (solid lines) and anisotropic $A_{1w3}^{(2)}$ (dotted lines) cylindrical Hamaker coefficients for the interaction of silica nanorods with biological materials in the fully retarded formulation with a screened zero-frequency contribution in the case of an intervening aqueous medium.

429 opposite to that of attractive interaction cases. If for illustration
430 purposes we compare the symmetric system (9,3,m)- Al_2O_3 -
431 (9,3,m) with asymmetric system (9,3,m)- Al_2O_3 -GC(10), we see
432 that indeed the torque as well as the force change sign. This
433 means that in the first case the two cylinders would tend to
434 align whereas in the second case they will tend to assume a
435 perpendicular configuration.

436 **3.5. Nonmonotonicity.** Nonmonotonic Hamaker coef-
437 ficients may be observed in asymmetric systems.

438 When the dielectric functions in a planar system change in a
439 stepwise manner, i.e., $\epsilon(i\xi)_1 > \epsilon(i\xi)_w > \epsilon(i\xi)_3$, the contribution
440 at those Matsubara frequencies $i\xi$ where the stepwise condition
441 is met adds a repulsive element to the overall vdW interaction.
442 This can lead to a range of effects including nonmonotonic
443 values of $A_{1w3}^{(0)}$ leading to a minimum in the interaction
444 energy or even overall net repulsive interactions.⁴⁷ In cylindrical
445 systems with anisotropic dielectric responses, however, this
446 condition is significantly more complicated because of the
447 intertwined morphological and optical anisotropy of the
448 cylinders, the latter a consequence of different axial and radial
449 components of the dielectric responses.³² Nevertheless, similar
450 effects on the sign of the force may be observed as well as on
451 the sign of torque. The Hamaker coefficients for a (24,24,s)
452 SWCNT interacting with a range of different materials (Figure
453 5) show a nonmonotonic dependence on spacing l for all
454 asymmetric systems shown here, wherein $A_{1w3}^{(0)}$ first rises
455 before falling again. This complex dependence may be readily
456 explained by observing the nature of the optical contrast in the
457 system. For the ((24,24,s)|w|collagen) system, the optical
458 properties $\epsilon(i\xi)$ (Figure 1, optical properties inset) show a clear
459 stepwise nature at energies above 10 eV, leading to repulsive
460 contributions from the high-energy Matsubara frequencies. As
461 these energies are gradually screened out by retardation, they
462 no longer contribute repulsive terms to the interaction, leading
463 to a net increase in $A_{1w3}^{(0)}$. At sufficiently large spacings, even
464 the attractive low-energy contributions are damped by
465 retardation, implying a decrease in $A_{1w3}^{(0)}$ as spacing tends
466 towards infinity. The asymmetric Hamaker coefficients, though
467 small, show an even more complex dependence on the
468 separation, with $A_{1w3}^{(2)}$ showing the same small-separation

469 local maximum, followed by another gradual rise at larger
470 separations. Under certain circumstances, including the
471 (24,24,s)|w|(Al₂O₃) system (green), $A_{1w3}^{(2)}$ shows a negative-
472 to-positive transition, indicating that increasing alignment
473 between the nanotubes may either increase or decrease the
474 interaction strength depending on the cylinder-cylinder
475 separation. This effect has never been observed or discussed
476 in other systems and would not be predicted by more simplistic
477 approaches to vdW interactions that ignore either optical
478 anisotropy or retardation effects.

479 **3.6. Hamaker Coefficient Data Mining.** The true
480 versatility of the *Gecko Hamaker* software platform implement-
481 ing the Lifshitz theory of vdW interactions is particularly
482 evident when considering systems with a wide range of
483 materials with disparate optical properties. Instead of
484 calculating the interaction characteristics of a single material
485 pair, the software facilitates the exploration of arrays of systems
486 to seek out the desired properties in order to guide material
487 selection and configuration (Figure 6). A number of trends are
488 immediately discernible. Torques and normal forces, which
489 depend on both Hamaker coefficients, show a strong
490 dependence on the cylinder radius as well as on the details of
491 the spectral properties of the interacting materials (Figure 6,
492 inset). This is evident in the case of interactions involving the
493 (24,24,s) SWCNT that show large torques and normal forces in
494 spite of their modest Hamaker coefficients, whereas small-
495 radius SWCNTs exhibit large values of $A_{1w3}^{(0)}$ and $A_{1w3}^{(2)}$ but
496 smaller torques. Nevertheless, in all systems investigated there
497 exists a strong normal force and a modest but always present
498 torque seeking to align the cylinders. Al_2O_3 interactions with
499 other inorganics tend to have disproportionately large torques
500 despite their relatively modest Hamaker coefficients and radii.
501 Through data mining the Hamaker coefficients, the full power
502 of *Gecko Hamaker* as a cross-system design platform comes to
503 the forefront: by analyzing across many different materials, one
504 may seek the interaction properties that are desired and
505 subsequently will be inspired to use materials that had not been
506 previously considered. Despite the power and versatility of the
507 *Gecko Hamaker* software platform, user interpretation of the
508 results is paramount. Hamaker coefficients express the material-

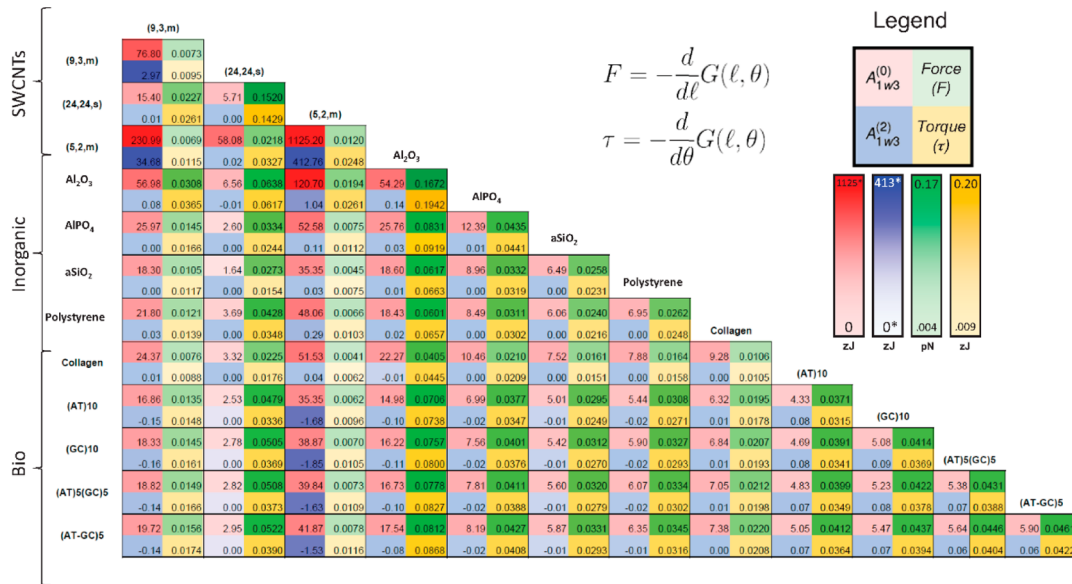


Figure 6. Pairs plot of symmetric and asymmetric interactions of anisotropic cylinders across an aqueous medium. The Hamaker coefficients, $A_{1w3}^{(0)}$ and $A_{1w3}^{(2)}$, and the forces and torques at a skew angle of 45° , at $l = 5$ nm separation are shown for interactions between a wide array of different materials, including biological materials, canonical inorganic materials, and SWCNTs. Cells are shaded according to interaction energy magnitudes. Torques and forces are calculated using commonly accepted literature values for radii of SWCNTs and biomaterials and a 1 nm radius for all other materials. The asterisk indicates that the color scale for $A_{1w3}^{(0)}$ saturates at 121 zJ for clarity and for $A_{1w3}^{(2)}$ saturates at 3.0 zJ, with negative values shaded according to their absolute value.

509 dependent nuances of the vdW interaction, but they are not
 510 equivalent to the interaction free energy itself, which has a
 511 separate dependence on cylinder radii, skew angle, and
 512 separation of its own. Effects such as retardation are therefore
 513 superimposed on these trends in the separation dependence of
 514 $A_{1w3}^{(0)}$ and $A_{1w3}^{(2)}$. As such, even when $A_{1w3}^{(2)}$ increases with
 515 separation, the net free energy of interaction may nonetheless
 516 monotonically decrease. It is important that these features of
 517 the Hamaker coefficients be properly interpreted and used for
 518 each particular application.

4. CONCLUSIONS

519 The mesoscale Lifshitz theory of vdW interactions has reached
 520 a level of refinement where it allows consistent and complete
 521 inclusion of solvent and temperature effects, retardation, optical
 522 anisotropy, morphology of interacting bodies, and geometry of
 523 interaction. In the limit of infinite area planar and infinitely long
 524 cylinder interaction geometries, it yields a full numerical
 525 implementation encoded in the *Gecko Hamaker* open-science
 526 software tool. The *Gecko Hamaker* software tool integrates a
 527 web service for the distribution of detailed optical properties of
 528 a broad range of heterogeneous functional materials from the
 529 spectral database and the newly implemented analytical
 530 solutions to the Lifshitz theory for a range of isotropic and
 531 anisotropic system configurations, yielding Hamaker coeffi-
 532 cients along with torques, forces, and free energies (Figure 1).
 533 The calculated Hamaker coefficients for cylindrical interaction
 534 geometry may vary by a few orders of magnitude, depending on
 535 the materials involved in the systems, with inorganic materials
 536 generally having larger Hamaker coefficients and biomolecular
 537 materials exhibiting smaller ones. The angular dependence of
 538 the vdW interaction, depending directly on the shape
 539 anisotropy and indirectly on the optical property anisotropy
 540 through the anisotropic part of the Hamaker coefficient, also
 541 varies significantly between materials. Both $A_{1w3}^{(0)}$ and $A_{1w3}^{(2)}$
 542 in fact exhibit dramatic and nonmonotonic retardation effects,

543 which vary significantly depending on the high-frequency
 544 contributions to the optical mismatch in a given system and are
 545 responsible for a wide array of interesting and unexpected
 546 effects. In addition, asymmetric systems may exhibit non-
 547 monotonic Hamaker coefficients, with distinct characteristics
 548 leading to novel and controllable design paradigms. What is
 549 important is the interconnectedness of all of these effects that
 550 precludes simple approximation schemes that focus on one or
 551 the other but misses the important links between them. The
 552 present mesoscale formulation of the classical Lifshitz theory of
 553 vdW interactions, taking fully into account the retardation as
 554 well as the anisotropy of interacting materials, constitutes a
 555 significant advance in accuracy and predictive power for the
 556 computation of vdW interactions compared to previous
 557 implementations of nonretarded, isotropic solutions. Further-
 558 more, the calculations accessible via the *Gecko Hamaker* open-
 559 science software tool may provide useful guidance to
 560 application engineers in streamlining the process of calculating
 561 appropriate Hamaker coefficients, eliminating the need for
 562 rough and misleading approximations, and making available
 563 optical properties of a wide variety of functional materials. This
 564 will allow a detailed investigation of the long-range interactions
 565 between materials and the design of specific model systems.

AUTHOR INFORMATION

Corresponding Author

*E-mail: rxf131@case.edu.

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Notes

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584 ■ REFERENCES

- 585 (1) Ninham, B. W.; Nostro, P. L. *Molecular Forces and Self Assembly*;
586 Cambridge University Press, 2011.
- 587 (2) Min, Y.; Akbulut, M.; Kristiansen, K.; Golan, Y.; Israelachvili, J.
588 The Role of Interparticle and External Forces in Nanoparticle
589 Assembly. *Nat. Mater.* **2013**, *7*, 527–538.
- 590 (3) Bishop, K. J. M.; Wilmer, C. E.; Soh, S.; Grzybowski, B. A.
591 Nanoscale Forces and Their Uses in Self-Assembly. *Small* **2009**, *5*,
592 1600–1630.
- 593 (4) Frenkel, D.; Wales, D. J. Colloidal Self-Assembly: Designed to
594 Yield. *Nat. Mater.* **2011**, *10*, 410–411.
- 595 (5) Nel, A. E.; Maedler, L.; Velegol, D.; Xia, T.; Hoek, E. M. V.;
596 Somasundaran, P.; Klaessig, F.; Castranova, V.; Thompson, M.
597 Understanding Biophysicochemical Interactions at the Nanobio
598 Interface. *Nat. Mater.* **2009**, *8*, 543–557.
- 599 (6) French, R.; Parsegian, V.; Podgornik, R.; Rajter, R.; Jagota, A.;
600 Luo, J.; Asthagiri, D.; Chaudhury, M.; Chiang, Y.; Granick, S.; et al.
601 Long Range Interactions in Nanoscale Science. *Rev. Mod. Phys.* **2010**,
602 *82*, 1887–1944.
- 603 (7) Drosdoff, D.; Woods, L. M. Quantum and Thermal Dispersion
604 Forces: Application to Graphene Nanoribbons. *Phys. Rev. Lett.* **2014**,
605 *112*, 025501.
- 606 (8) Rajter, R. F.; French, R. H.; Ching, W. Y.; Podgornik, R.;
607 Parsegian, V. A. Chirality-Dependent Properties of Carbon Nano-
608 tubes: Electronic Structure, Optical Dispersion Properties, Hamaker
609 Coefficients and van Der Waals-London Dispersion Interactions. *RSC*
610 *Adv.* **2013**, *3*, 823–842.
- 611 (9) Shih, C.-J.; Strano, M. S.; Blankschtein, D. Wetting Translucency
612 of Graphene. *Nat. Mater.* **2013**, *12*, 866–869.
- 613 (10) Geim, A. K.; Grigorieva, I. V. Van Der Waals Heterostructures.
614 *Nature* **2011**, *479*, 419–425.
- 615 (11) Parsegian, V. A. *Van Der Waals Forces: A Handbook for Biologists*,
616 *Chemists, Engineers, and Physicists*; Cambridge University Press, 2006.
- 617 (12) Klimes, J.; Michaelides, A. Perspective: Advances and
618 Challenges in Treating van Der Waals Dispersion Forces in Density
619 Functional Theory. *J. Chem. Phys.* **2012**, *137*, 120901–120912.
- 620 (13) Emig, T. Casimir Physics: Geometry, Shape and Material. *J.*
621 *Mod. Phys. A* **2010**, *25*, 2177–2195.
- 622 (14) Dalvit, D.; Milonni, P.; Roberts, D.; Rosa, F. *Casimir Physics*;
623 Springer, 2011.
- 624 (15) Rodriguez, A. W.; Capasso, F.; Johnson, S. G. The Casimir
625 Effect in Microstructured Geometries. *Nat. Photonics* **2011**, *5*, 211–
626 221.
- 627 (16) Gobre, V. V.; Tkatchenko, A. Scaling Laws for van Der Waals
628 Interactions in Nanostructured Materials. *Nat. Commun.* **2013**, *4*,
629 2341–2346.
- 630 (17) French, R. H. Origins and Applications of London Dispersion
631 Forces and Hamaker Constants in Ceramics. *J. Am. Ceram. Soc.* **2000**,
632 *83*, 2117–2146.
- 633 (18) Hopkins, J. C.; Dryden, D. M.; Ching, W.-Y.; French, R. H.;
634 Parsegian, V. A.; Podgornik, R. Dielectric Response Variation and the
635 Strength of van Der Waals Interactions. *J. Colloid Interface Sci.* **2014**,
636 *417*, 278–284.
- 637 (19) Munday, J. N.; Iannuzzi, D.; Barash, Y.; Capasso, F. Torque on
638 Birefringent Plates Induced by Quantum Fluctuations. *Phys. Rev. A*
639 **2005**, *71*, 042102.
- (20) Narayanaswamy, A.; Zheng, Y. Van Der Waals Energy and
640 Pressure in Dissipative Media: Fluctuational Electrodynamics and
641 Mode Summation. *Phys. Rev. A* **2013**, *88*, 012502.
- (21) Warner, J. H.; Young, N. P.; Kirkland, A. I.; Briggs, G. A. D.
642 Resolving Strain in Carbon Nanotubes at the Atomic Level. *Nat.*
643 *Mater.* **2011**, *10*, 958–962.
- (22) Strano, M. S.; Jain, R. M.; Tvrdy, K.; Han, R.; Ulissi, Z. W. A
644 Quantitative Theory of Adsorptive Separation for the Electronic
645 Sorting of Single-Walled Carbon Nanotubes. *ACS Nano* **2014**, 3367–
646 3379.
- (23) Srivastava, A.; Srivastava, O. N.; Talapatra, S.; Vajtai, R.; Ajayan,
647 P. M. Carbon Nanotube Filters. *Nat. Mater.* **2004**, *3*, 610–614.
- (24) Sethi, S.; Ge, L.; Ci, L.; Ajayan, P. M.; Dhinojwala, A. Gecko-
648 Inspired Carbon Nanotube-Based Self-Cleaning Adhesives. *Nano Lett.*
649 **2008**, *8*, 822–825.
- (25) Zheng, J.; Birktoft, J. J.; Chen, Y.; Wang, T.; Sha, R.;
650 Constantinou, P. E.; Ginell, S. L.; Mao, C.; Seeman, N. C. From
651 Molecular to Macroscopic via the Rational Design of a Self-Assembled
652 3D DNA Crystal. *Nature* **2009**, *461*, 74–77.
- (26) Young, K. L.; Ross, M. B.; Blaber, M. G.; Rycenga, M.; Jones, M.
653 R.; Zhang, C.; Senesi, A. J.; Lee, B.; Schatz, G. C.; Mirkin, C. A. Using
654 DNA to Design Plasmonic Metamaterials with Tunable Optical
655 Properties. *Adv. Mater.* **2014**, *26*, 653–659.
- (27) Liu, N.; Hentschel, M.; Weiss, T.; Alivisatos, A. P.; Giessen, H.
656 Three-Dimensional Plasmon Rulers. *Science* **2011**, *332*, 1407–1410.
- (28) Cheng, X.; Gurkan, U. A.; Dehen, C. J.; Tate, M. P.; Hillhouse,
657 H. W.; Simpson, G. J.; Akkus, O. An Electrochemical Fabrication
658 Process for the Assembly of Anisotropically Oriented Collagen
659 Bundles. *Biomaterials* **2008**, *29*, 3278–3288.
- (29) Jin, M.; Feng, X.; Feng, L.; Sun, T.; Zhai, J.; Li, T.; Jiang, L.
660 Superhydrophobic Aligned Polystyrene Nanotube Films with High
661 Adhesive Force. *Adv. Mater.* **2005**, *17*, 1977–1981.
- (30) Capasso, F.; Munday, J. N.; Iannuzzi, D.; Chan, H. B. Casimir
662 Forces and Quantum Electrodynamical Torques: Physics and
663 Nanomechanics. *IEEE J. Sel. Top. Quantum Electron.* **2007**, *13*, 400–
664 414.
- (31) *Gecko Hamaker*; 2014.
- (32) Siber, A.; Rajter, R.; French, R.; Ching, W.; Parsegian, V.;
665 Podgornik, R. Dispersion Interactions between Optically Anisotropic
666 Cylinders at All Separations: Retardation Effects for Insulating and
667 Semiconducting Single-Wall Carbon Nanotubes. *Phys. Rev. B* **2009**, *80*,
668 165414.
- (33) Pitaevskii, L. P. Thermal Lifshitz Force between an Atom and a
669 Conductor with a Small Density of Carriers. *Phys. Rev. Lett.* **2008**, *101*,
670 163202.
- (34) Dobson, J. F.; Gould, T. Calculation of Dispersion Energies. *J.*
671 *Phys.: Condens. Matter* **2012**, *24*, 073201.
- (35) Tkatchenko, A. Current Understanding of Van Der Waals
672 Effects in Realistic Materials. *Adv. Funct. Mater.* **2014**, n/a–n/a.
- (36) Jr, R. A. D.; Gobre, V. V.; Tkatchenko, A. Many-Body van Der
673 Waals Interactions in Molecules and Condensed Matter. *J. Phys.:*
674 *Condens. Matter* **2014**, *26*, 213202.
- (37) Zhao, R.; Luo, Y.; Fernández-Domínguez, A. I.; Pendry, J. B.
675 Description of van Der Waals Interactions Using Transformation
676 Optics. *Phys. Rev. Lett.* **2013**, *111*, 033602.
- (38) Argento, C.; Jagota, A.; Carter, W. C. Surface Formulation for
677 Molecular Interactions of Macroscopic Bodies. *J. Mech. Phys. Solids*
678 **1997**, *45*, 1161–1183.
- (39) Van Benthem, K.; Tan, G.; French, R. H.; DeNoyer, L. K.;
679 Podgornik, R.; Parsegian, V. A. Graded Interface Models for More
680 Accurate Determination of van Der Waals–London Dispersion
681 Interactions across Grain Boundaries. *Phys. Rev. B* **2006**, *74*, 205110.
- (40) Poudel, L.; Rulis, P.; Liang, L.; Ching, W.-Y. Partial Charge and
682 Hydrogen Bonding in Four Periodic B-DNA Models. *Phys. Rev. E*
683 **2014**, 2705–2705.
- (41) French, R.; Mullejans, H.; Jones, D.; Duscher, G.; Cannon, R.;
684 Ruhle, M. Dispersion Forces and Hamaker Constants for Intergranular
685 Films in Silicon Nitride from Spatially Resolved-Valence Electron
686 Energy Loss Spectrum Imaging. *Acta Mater.* **1998**, *46*, 2271–2287.

- 709 (42) Ching, W.-Y.; Rulis, P. *Electronic Structure Methods for Complex*
710 *Materials: The Orthogonalized Linear Combination of Atomic Orbitals*;
711 Oxford University Press, 2012.
- 712 (43) Holdren, J. P. *Increasing Access to the Results of Federally Funded*
713 *Scientific Research*; Executive Office of the President: Office of Science
714 and Technology Policy, 2013.
- 715 (44) Obama B. H. *Executive Order – Making Open and Machine*
716 *Readable the New Default for Government Information*; 2013.
- 717 (45) Open Data Charter; [https://www.gov.uk/government/](https://www.gov.uk/government/publications/open-Data-Charter)
718 [publications/open-Data-Charter](https://www.gov.uk/government/publications/open-Data-Charter), 2013.
- 719 (46) Homer Reid M. T.; Johnson S. G. Efficient Computation of Power,
720 Force, and Torque in BEM Scattering Calculations. *arXiv:1307.2966*,
721 **2013**.
- 722 (47) Elbaum, M.; Schick, M. Application of the Theory of Dispersion
723 Forces to the Surface Melting of Ice. *Phys. Rev. Lett.* **1991**, *66*, 1713–
724 1716.
- 725 (48) Zhang, G.-J.; Zhang, G.; Chua, J. H.; Chee, R.-E.; Wong, E. H.;
726 Agarwal, A.; Buddharaju, K. D.; Singh, N.; Gao, Z.; Balasubramanian,
727 N. DNA Sensing by Silicon Nanowire: Charge Layer Distance
728 Dependence. *Nano Lett.* **2008**, *8*, 1066–1070.